

## Article

# Achieving Ultralow-Density, High-Purity Au Foam Hohlräum with Hierarchical Porous Structure

Xiulan Tan <sup>1,†</sup>, Jian Wang <sup>2,†</sup>, Jiangshan Luo <sup>1</sup>, Gao Niu <sup>1,\*</sup> and Chaoyang Wang <sup>1</sup>

<sup>1</sup> Research Center of Laser Fusion, China Academy of Engineering Physics, Mianyang 621900, China

<sup>2</sup> Joint Laboratory for Extreme Conditions Matter Properties, Southwest University of Science and Technology, Mianyang 621010, China

\* Correspondence: niugao@caep.cn

† These authors contributed equally to this work.

**Abstract:** High Z (atomic number) metallic foams with a low density and high purity are urgent demands in high energy-density physical experiments. They suppress plasma expansion and convert the laser pulses to X-rays more uniformly and efficiently. Thus, we synthesized an ultralow-density and high-purity Au foam hohlraum with a hierarchical porous structure via a template-dealloying method in this paper. Silica (SiO<sub>2</sub>) beads were introduced as the sacrificial templates due to their high stability at an elevated temperature. The Au and Ag nanoparticles were successively deposited onto the SiO<sub>2</sub> templates via an electroless deposition process to form an Ag@Au@SiO<sub>2</sub> core-shell structure. Cylindrical Ag@Au@SiO<sub>2</sub> hohlraum was achieved using a filter-casting technique with a patented mold. Afterward, an Au-Ag alloy was generated during 36 h of calcination at 400 °C. Self-supported Au foam hohlraum with the hierarchical porous structure was gained after the SiO<sub>2</sub> templates were removed, followed by the dealloying of the Ag from the Au-Ag alloy. A self-supporting Au foam hohlraum with a density as low as 0.2 g/cm<sup>3</sup> and a purity of 99.37% was achieved, and the density decreased by about 44.5% when compared with our previous Au foam (density: 0.36 g/cm<sup>3</sup>, purity: less than 96%) using microspherical polystyrene as the sacrificial template. Thus, the ultralow-density, high-purity Au foam hohlraum may exhibit profound application in high-energy physical experiments in the near future.

**Keywords:** template-dealloying; Au foam; ultralow-density; inertial confinement fusion



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## 1. Introduction

Low-density, high-purity metallic foams with hierarchical porous structures have attracted tremendous interest in academic and industrial circles [1–4]. Firstly, the metallic mainframe provides the material with high electrical, excellent heat conductivity, and high strength [3]; additionally, the hierarchical porous structure with high porosity has an enhanced specific area to absorb more ions (or molecules) and facilitate mass transportation [5,6]. Thus, they are widely used in catalytic, electrochemical, sensor, and purification [7–15]. Synthesis of porous noble metal materials with multimodal structures is still challenging, although significant progress in fabricating multimodal porous ceramics, metal oxides, and organic carbon has been achieved.

The exploitation of nuclear fusion, which generates endless energy without any carbon dioxide or wastes from the reaction process, has been considered one of the promising solutions to the global energy crisis. At present, two possible methods are ongoing, one is the international thermonuclear experimental reactor (ITER) project and another way to obtain a controlled fusion reaction is inertial confinement fusion (ICF) with high-power lasers [16]. In the indirect-drive ICF, a high Z (high atomic number, high density, and high radiation opacity) metal hohlraum, which converts the laser pulses into X-rays, is indispensable [17]. The available X-ray radiation amount depends on the conversion efficiency of the laser to X-ray and the X-ray reemission efficiency of the hohlraum [18,19].

According to the theoretical and experimental calculations, the foam hohlraum with lower density and higher purity features lower X-ray energy loss with higher conversion efficiency and obtains a higher radiation temperature [19–22]. Thus, as the frequently utilized high Z metal, the Au foam having a desirable density, purity, and porosity, is a promising candidate for hohlraum wall materials in ICF facilities [23,24].

Template-assisted synthesis and dealloying are the two main methods used to form pores in metals [25]. Templating uses sacrificial organic or inorganic templates to construct pores, and these pore sizes are template-dependent, ranging from a few nanometers to tens of micrometers [26–28]. While in the dealloying process, the least noble component in an alloy is selectively removed by corrosion, leaving the more noble element forming a nanoporous structure [7,29,30]. The pore size distribution can only be adjusted in a narrow range by controlling the ratios of the components in the alloy. However, obtaining porous metals with an ultralow density by a single method is hard to achieve. Thus, combining the templating and dealloying techniques could fabricate multimodal porous noble metal materials [31].

Herein, a template-dealloying strategy to fabricate ultralow-density, high-purity gold foam hohlraum with a hierarchical porous structure was demonstrated, which consisted of multiple steps. The monodispersed silica ( $\text{SiO}_2$ ) beads preseeded with Au nanoparticles were selected as the sacrificial templates, followed by Au and Ag electroless deposition, and an  $\text{Ag@Au@SiO}_2$  core-shell structure was obtained. Subsequently, the Ag and Au were alloyed at a constant temperature for tens of hours, followed by removing the  $\text{SiO}_2$  templates and Ag component corrosion, and an Au foam hohlraum with a hierarchical porous structure was gained.

## 2. Experimental Section

### 2.1. Materials

The monodispersed  $\text{SiO}_2$  beads (10  $\mu\text{m}$ ), chloroauric acid ( $\text{HAuCl}_4 \cdot 4\text{H}_2\text{O}$ ), silver nitrate ( $\text{AgNO}_3$ ), tetraoctyl ammonium bromide, and hydroxylamine hydrochloride were obtained from Aladdin.

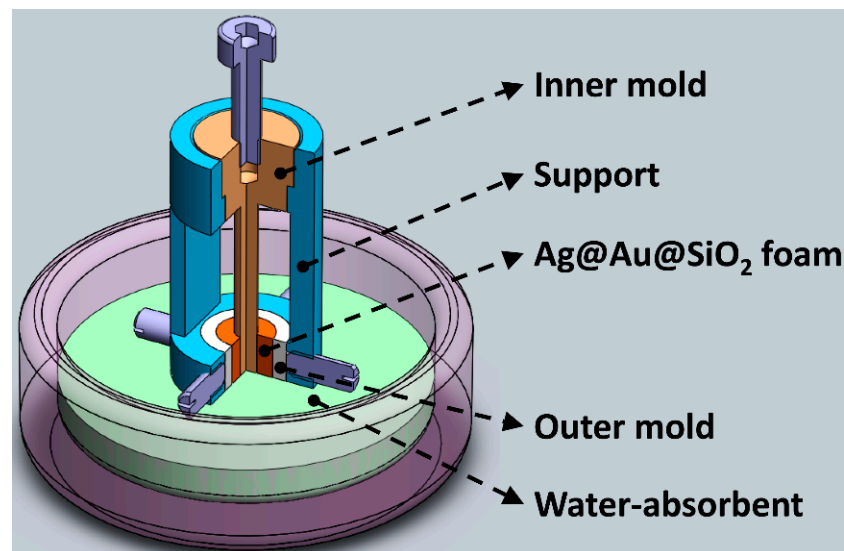
### 2.2. Synthesis of Au Foam Hohlraum

The preparation of low-density Au foam hohlraum is multistep works, as schematically shown in Figure 1. Step 1, the Au nanoparticle colloid, which served as the nucleation site for the subsequent electroless Au deposition, was prepared according to the literature [32]. The monodisperse  $\text{SiO}_2$  beads were modified by the 3-aminopropyltriethoxysilane, resulting in positively charged  $\text{SiO}_2$  beads grafted by the amino groups. Step 2, the Au nanoparticles firmly covered the  $\text{SiO}_2$  beads by dropwise adding the Au nanoparticle colloid into the  $\text{SiO}_2$  beads solution and sufficiently stirring. After washing and centrifugation, the Au nanoparticles coated  $\text{SiO}_2$  beads were re-dispersed in 80 mL of deionized water. Step 3, a thicker Au nanoparticles layer was obtained by an electroless deposition process, in which the polyvinylpyrrolidone (PVP), hydroxylamine hydrochloride, and  $\text{HAuCl}_4$  were used as a stabilizing agent, reduction agent, and an Au source, respectively. Step 4, after careful washing and centrifugation, the sample was re-dispersed in 80 mL of deionized water for Ag electroless deposition. During the Ag deposition process, the PVP was introduced as a stabilizer, aqueous ammonia and glucose as reductants, and  $\text{AgNO}_3$  as an Ag source, respectively. Step 5, the  $\text{Ag@Au@SiO}_2$  beads were dispersed in 2 mL of ionized water after the as-received solution was washed and centrifuged. A free-standing hollow cylinder-shaped  $\text{Ag@Au@SiO}_2$  foam was formed by a filter-casting method using a patented mold (Figure 2). Step 6, after drying and demolding, the  $\text{Ag@Au@SiO}_2$  foam was calcined at 400  $^\circ\text{C}$  for 36 h in a flowing nitrogen atmosphere, forming an Au-Ag alloy on the outside surface of the  $\text{SiO}_2$  beads. Step 7, the  $\text{SiO}_2$  bead templates were removed in a hydrofluoric acid solution by gradually increasing the concentrations (1%, 4%, and 10%). Step 8, the less noble content of Ag in the Au-Ag alloy was selectively removed via a dealloying process, which was performed in a nitric acid solution by gradually increasing the

concentration (3.55 M, 7.9 M, 11.9 M, and concentrated nitric acid). The Au foam hohlraum was washed several times with deionized water to remove the residual nitric acid and then placed in acetone to completely displace the water. Finally, an ultralow-density, high purity, and hierarchical porous Au foam hohlraum was obtained after drying in a supercritical CO<sub>2</sub> atmosphere to prevent the Au foam hohlraum from shrinkage.



**Figure 1.** Schematic illustration of Au foam hohlraum fabrication.



**Figure 2.** The structure diagram of the patented mold used to fabricate Ag@Au@SiO<sub>2</sub> hohlraum.

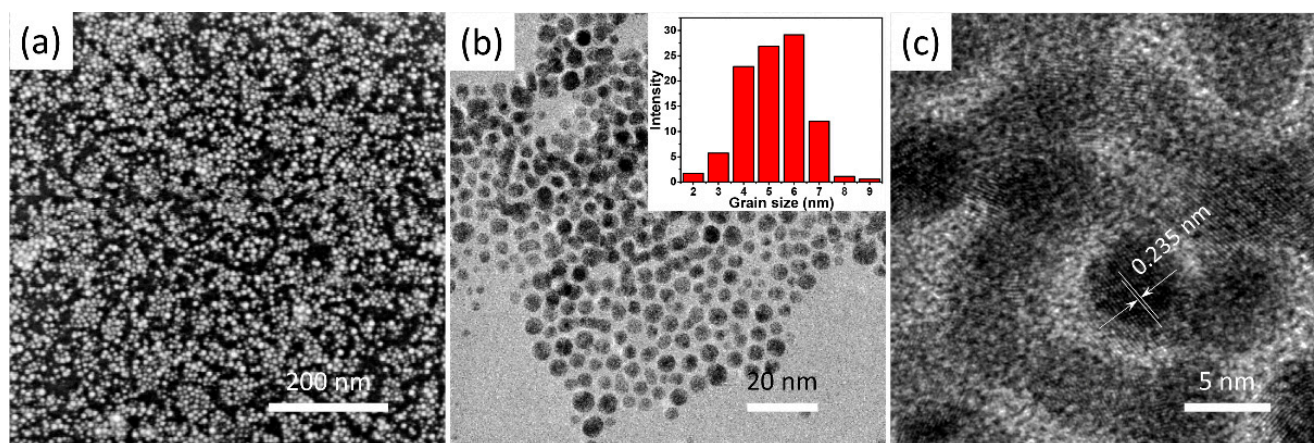
### 2.3. Characterization

The morphology of the as-received Au foams was investigated by a field-emission scanning electron microscope (FESEM), and the elemental compositions were investigated by energy dispersive spectrum (EDS), which is attached to the FESEM system. Meanwhile, a focus ion beam (FIB) was employed to slice the Au foam hohlraum and the thickness was observed by the FESEM. The densities of the Au foam hohlraum were obtained by dividing the mass of the Au foam hohlraum by the volume. The contents of impurities were analyzed by inductively coupled plasma mass spectrum (ICP-MS), and the carbon and oxygen contents were determined by a carbon analyzer. The ligament size distribution of the Au foams was calculated using an image analysis software (Image-Pro Plus) from the FESEM images. The N<sub>2</sub> adsorption/desorption isotherms were conducted with a surface area analyzer at 196 °C. The specific surface and the pore distribution of the porous

Au foams were estimated by the Brunauer-Emmett-Teller (BET) method and the density functional theory (DFT) program, respectively.

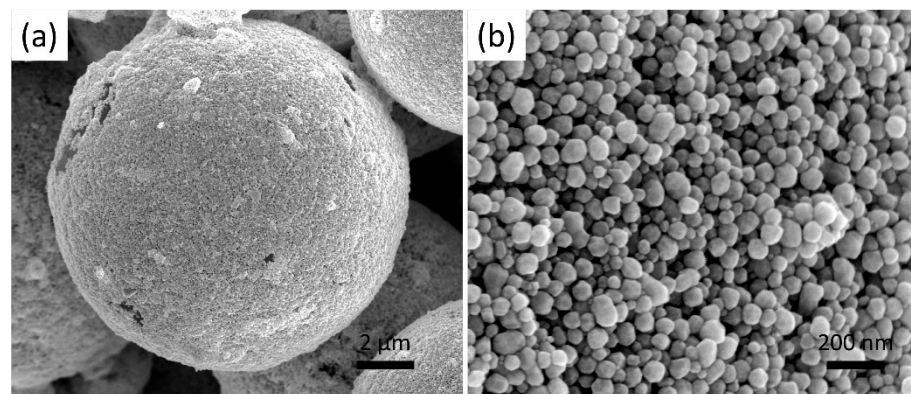
### 3. Results and Discussion

Firstly, the Au nanoparticle colloid used as the nucleation sites in the following Au electroless deposition process was synthesized using a spontaneous phase transfer method, which transfers the Au nanoparticles from the toluene to deionized water by the addition of 4-dimethylaminopyridine [32]. The as-received Au nanoparticle colloid exhibits excellent stability and dispersion. Moreover, the Au nanoparticles in the colloid have a uniform size distribution with an average diameter of 5.7 nm, according to the SEM and TEM analysis (Figure 3), which is beneficial for the subsequent Au electroless deposition. The high-resolution TEM image in Figure 3c exhibits clear lattice fringes with an interplanar space of 0.235 nm, which can be well indexed to the Au (111) plane, confirming that the Au nanoparticles used in the phase transfer method are in a good crystalline structure.

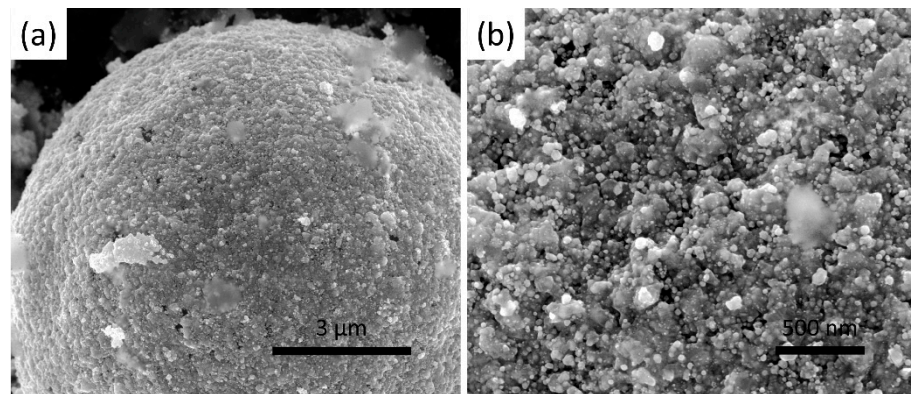


**Figure 3.** (a) high magnification SEM image, (b) TEM image, and (c) high magnification TEM image of the prepared Au nanoparticles, the inset of (b) shows the size distribution of the prepared Au nanoparticles.

The Ag@Au@SiO<sub>2</sub> core-shell microstructure was prepared by electroless deposition. During the Au electroless deposition, the pre-seeded Au nanoparticles on the SiO<sub>2</sub> beads grew further, and the deposited Au nanoparticles layer thickened, as the hydroxylamine hydrochloride gradually reduced the HAuCl<sub>4</sub> into Au nanoparticles. The deposited Au nanoparticles exhibit a uniform spherical morphology with a grain size of a few tens to a hundred nm (Figure 4) and without the observation of any aggregations. The PVP molecules, with the macromolecule long chains, stretch out and twine around the Au nanoparticles, preventing the Au nanoparticles aggregation with the help of constant stirring [31]. Moreover, the pre-seeded Au nanoparticles with lower energy provide multiple initial nucleation sites for the subsequent Au electroless deposition with a high reaction rate. While during the Ag electroless deposition process, aqueous ammonia and glucose were used as the reductants to reduce the AgNO<sub>3</sub> into Ag nanoparticles which adhered to the deposited Au layer. The morphology of the synthesized Ag coating with different magnifications is shown in Figure 5, which differs from the Au nanoparticles via the electroless deposition process. The Ag nanoparticles gradually fill the gaps between the Au nanoparticles and form a compact layer. It may result that the gaps between the Au nanoparticles, with lower energy, are beneficial for Ag nucleation. Au electroless deposition must be carried out before Ag electroless deposition, because depositing Au on Ag may result in competing Ag<sup>0</sup>/Au<sup>3+</sup> galvanic and electroless deposition reduction reactions [33].



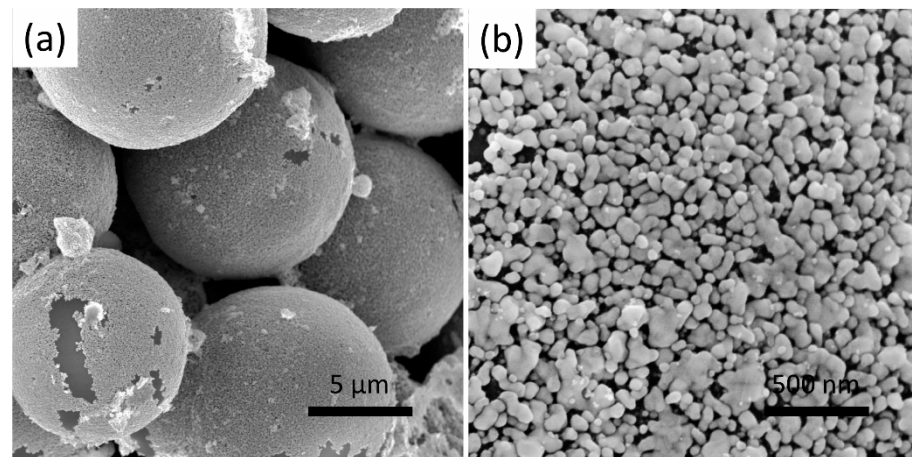
**Figure 4.** (a) low and (b) high magnification SEM images of SiO<sub>2</sub> beads after Au nanoparticles electroless deposition.



**Figure 5.** (a) low and (b) high magnification SEM images of SiO<sub>2</sub> beads after Ag nanoparticles electroless deposition.

As the electroless deposition of the Au and Ag was complete, the coated SiO<sub>2</sub> beads were collected, re-dispersed in distilled water, and then cast to construct an Ag@Au@SiO<sub>2</sub> hohlraum. Due to the Ag@Au@SiO<sub>2</sub> beads, prepared via the multi-steps, having good dispersity, thus, they can be easily dispersed in distilled water. A self-designed mold (Figure 2) was employed to cast the Au foam hohlraum. The mold contains four parts, a platform, a hollow support, a Teflon inner mold, and a Teflon outer mold. The platform, filled with compacted plaster of Paris (CaSO<sub>4</sub>), is used to timely absorb water, solidify the Ag@Au@SiO<sub>2</sub> foam, and support the whole mold during the casting process. The Teflon outer mold is fixed to the hollow support with four bolts, while the inner mold is placed in the guide hole of the hollow support. The Ag@Au@SiO<sub>2</sub> water suspension is dropped into the gap between the outer and inner Teflon mold and settled down gradually with a random arrangement. Thus, it is easy to conclude that the outer diameter of the inner Teflon mold and the inner diameter of the outer Teflon mold determine the inner and outer diameter of the Au hohlraum, and the height of the molds limit the height of the hohlraum. Furthermore, multiple series of inner and outer molds are designed to meet the diverse demands of the ICF experiments. Eventually, all the Ag@Au@SiO<sub>2</sub> beads settled down and a hohlraum, with the expected shape and size, formed. Demolding can be readily performed by disassembling the fixed bolts and pulling out the inner mold after desiccation at 40 °C in an incubator for half an hour. The fabricated hohlraum was calcinated in an N<sub>2</sub> atmosphere at a fixed temperature of 400 °C for 36 h after demolding, and the alloying reaction of the Au and Ag took place during the heat treatment. Figure 6 shows the SEM images of the obtained Au-Ag alloy with different magnifications. The Au-Ag alloy is tightly adhered to the SiO<sub>2</sub> beads and exhibits a porous nanostructure with varied alloy nanoparticle grain sizes, the morphology of which is different from the

deposited Au nanoparticles and the coated Ag layer. The volume of the metal nanoparticles gradually shrinks to form a porous nanostructure when the alloying is processed at a relatively high temperature. Finally, an Au foam hohlraum, with the expected size and shape, was obtained by removing the sacrificial SiO<sub>2</sub> template and selectively etching the less noble Ag component in the alloy.

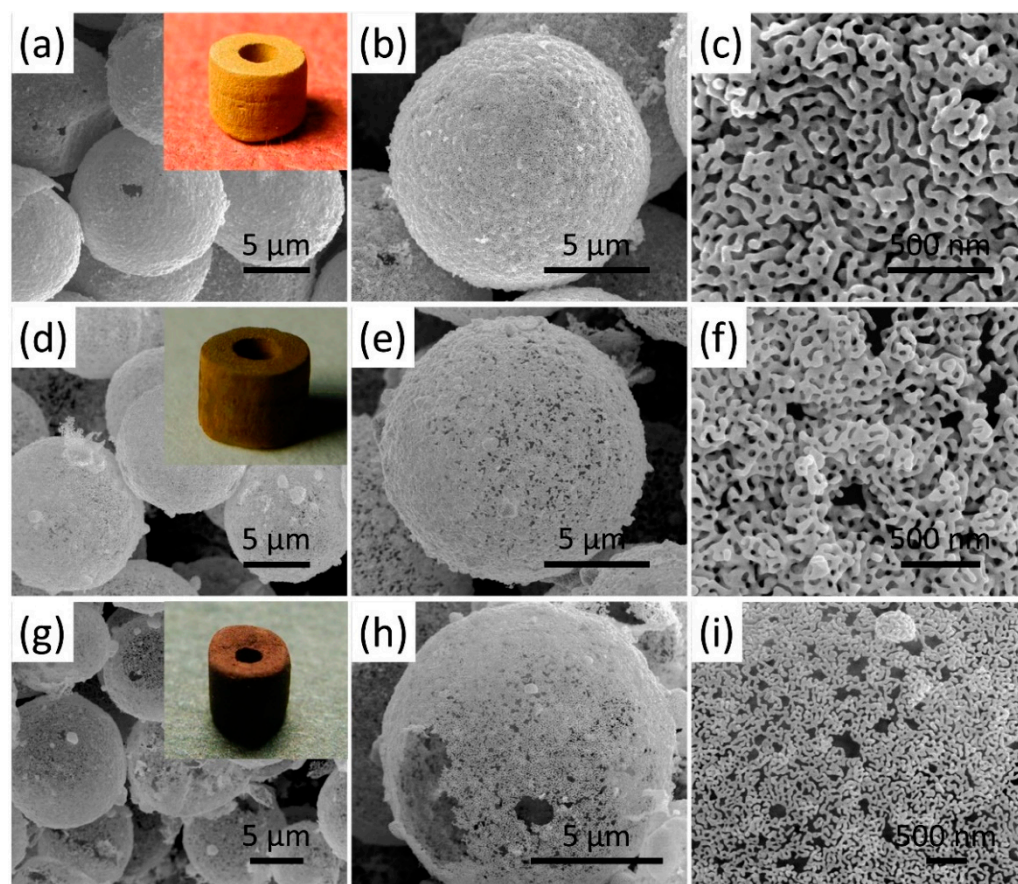


**Figure 6.** (a) low and (b) high magnification SEM images of Au-Ag alloy coated on SiO<sub>2</sub> beads after calcined at 400 °C for 36 h.

It is well known that the densities of the synthesized Au foam hohlraum can be facilely adjusted by tuning the diameter of the SiO<sub>2</sub> beads, varying the mole ratios of the deposited Au and Ag, as well as controlling the amount (thickness) of the coated Au and Ag during electroless deposition conditions. In our study, other sizes of SiO<sub>2</sub> bead were also selected to fabricate Au hohlraum, all exhibiting low density and high purity using the template-dealloying method, and the larger the SiO<sub>2</sub> bead, the lower density achieved. However, the thickness of the Au hohlraum is limited to 150~200 μm to meet the physical experiment design, and when the chosen SiO<sub>2</sub> bead is more than 10 μm, the strength and mechanical performance of the Au hohlraum will decrease. Thus, we selected the 10 μm SiO<sub>2</sub> bead as the template. In the present work, the Au foam hohlraum, with varied densities, was achieved by simply adjusting the mole ratios of the deposited Au and Ag layer with a fixed size of SiO<sub>2</sub> beads and Au amount.

Figure 7 shows the SEM images of the synthesized Au foam hohlraum with different densities by the template-dealloying method, in which SiO<sub>2</sub> beads (10 μm) as templates. As can be seen, the Au foam hohlraum with a hierarchical porous structure is composed of single self-supporting hollow Au shells stacked closely together. The pores mainly consist of three parts, the hollow shell structure after removing the SiO<sub>2</sub> templates, the porous nanostructure after selectively etching the less noble Ag component in the Au-Ag alloy, and the gaps between the randomly stacked shells. The insets of Figure 7a,d,g show the relevant digital images of the prepared Au foam hohlraum, and significant color changes are observed with the decrease of the densities, varying from golden to bronze and further to brown. The yellow feature of bulky gold comes from the complementary blue absorption by the interband transition. In Au foams, the free electrons of nanostructures interact with light, leading to localized surface plasma. The plasma absorbs the light out of the blue range. Thus, the Au foams' reflected spectrum and color change (Figure 8). In addition, when the Au density decreases, the nanostructures would be smaller and can trap more light in the entire spectrum, resulting in a darker foam. The Au foam hohlraum densities decrease as the thickness of the porous Au shell reduces. The thinner the Au shell, the lower density it has. Taking the Au foam hohlraum with a density of 0.41 g/cm<sup>3</sup> as an example, the thickness is 170~270 nm. The thickness of the Au shell decreases to 150~200 nm as the density reduces to 0.30 g/cm<sup>3</sup> and further becomes 76~140 nm as the density decreases to

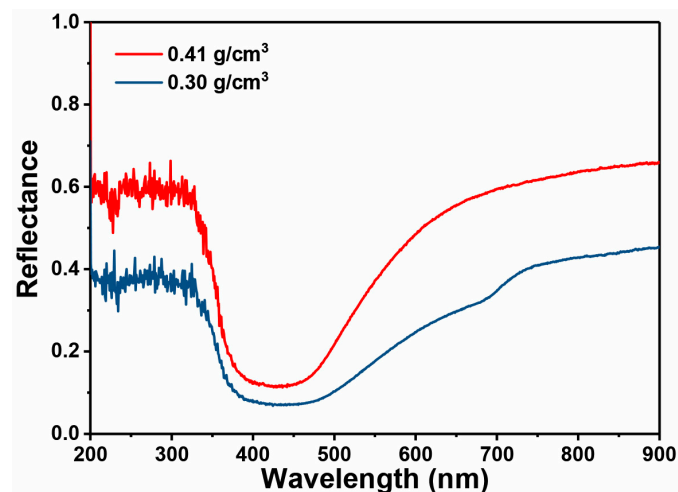
0.20 g/cm<sup>3</sup>. Furthermore, the ligament size distribution of the porous Au shell is closely relevant to the densities of Au foam hohlraum. For example, when the Au foam hohlraum has a density of 0.41 g/cm<sup>3</sup>, as high as 76% of ligament distributes in a narrow range of 45~55 nm. As the density of Au foam hohlraum comes to 0.30 g/cm<sup>3</sup>, the distribution range of ligament size gradually widens, from 35 to 65 nm, and the 45~55 nm ligament takes up to 57%. As the density of Au foam hohlraum further decreases to 0.20 g/cm<sup>3</sup>, the ligament size distribution further broadens to the range of 45~100 nm, and only 11% of the ligament is between 45 and 55 nm, while about 72% of the ligament distributes from 65 to 85 nm. What's more, it is observed from Figure 7i that the Au foam hohlraum with a density of 0.20 g/cm<sup>3</sup> almost consists of a single thin layer of nanoporous structure and further decreases the thickness of the plated layer will result in the crack or collapse of the Au shell (Figure 9). Thus, it is believed that the Au layer thickness with a density of 0.20 g/cm<sup>3</sup> is near the limit thickness, and the density is close to the limit density of Au foam hohlraum prepared by the template-dealloying method using the SiO<sub>2</sub> beads (10 µm) as the sacrificial template.



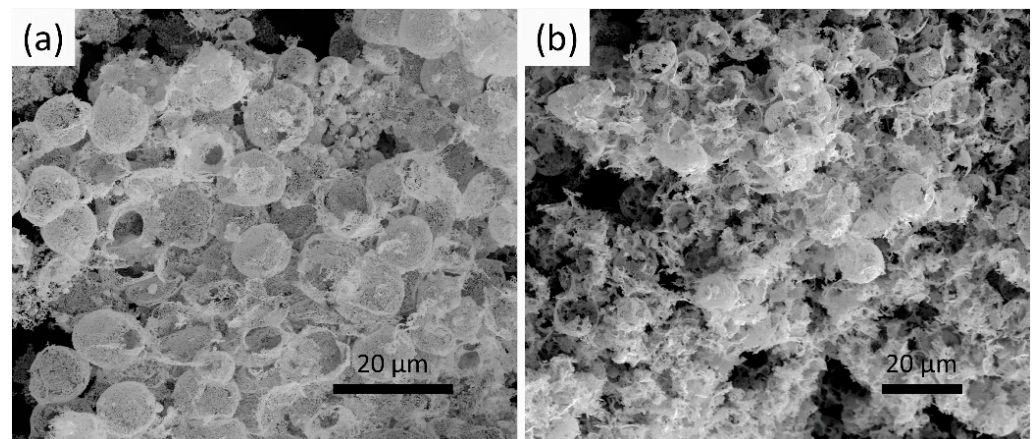
**Figure 7.** SEM images of Au foam hohlraum with a density of (a–c) 0.41, (d–f) 0.30, and (g–i) 0.20 g/cm<sup>3</sup> with different magnifications, the insets of (a,d,g) show the digital images of the corresponding Au foam hohlraum.

Apart from the hierarchical porous structure, another critical factor to ensure the Au foam hohlraum, with an ultralow density (as low as 0.20 g/cm<sup>3</sup>), is that the SiO<sub>2</sub> beads are employed as templates, the size of which hardly shrink during the whole preparation process. Previous studies have shown that the polystyrene (PS) beads can be used as templates for the low-density Au foam synthesis and they can be easily removed by a heat treatment process when the Au and Ag are alloying [31,34,35]. Unfortunately, the size of the PS bead shrinks obviously (as high as 20%) during the pyrolysis process, which impedes the density of the Au foam from further decreasing. Moreover, the plated Au-Ag layer re-

aggregates and regrows during heat treatment, showing a significant sintering morphology. However, it is far different from the situation with the PS beads as the template, where the  $\text{SiO}_2$  beads template will not remove during the alloying process, and no significant size shrinkage happens after the heat treatment. Additionally, only the re-aggregation and regrowth of the Au and Ag nanoparticles in the plated layer are detected during the alloying. Thus, when the PS and  $\text{SiO}_2$  beads are in the same diameter, and the ratio and amount of the gold and the silver are equal, a thinner Au shell can be achieved when using the  $\text{SiO}_2$  beads compared with the PS beads as a template.



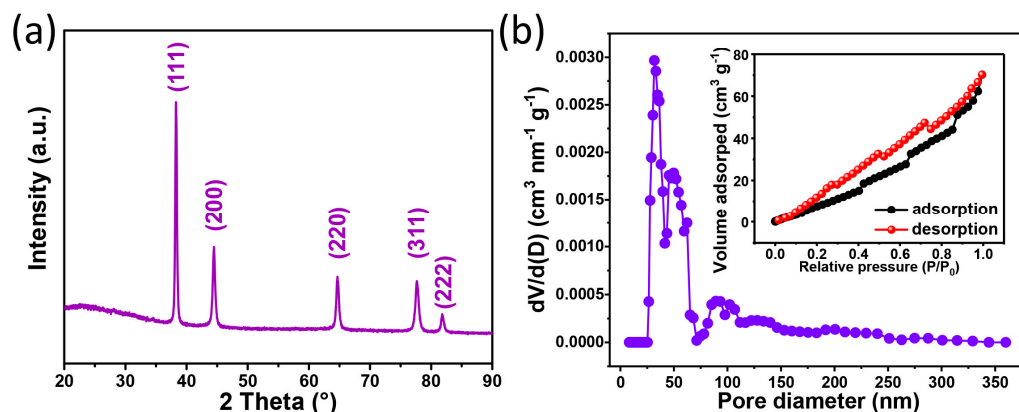
**Figure 8.** Reflection spectra of Au foam hohlraum with different densities of  $0.41 \text{ g/cm}^3$  and  $0.30 \text{ g/cm}^3$ .



**Figure 9.** SEM images of (a) the cracked and (b) collapse of the Au foam.

The Au foam hohlraum, with a density of  $0.20 \text{ g/cm}^3$ , was characterized by X-ray diffraction, and the result is presented in Figure 10a. Five diffraction peaks located at the  $2\theta$  of  $38.3^\circ$ ,  $44.6^\circ$ ,  $64.6^\circ$ ,  $77.5^\circ$ , and  $82.3^\circ$  with different intensities are detected and can be well indexed to the (111), (200), (220), (311), and (222) planes of gold (PDF No. 00-001-1172), respectively. The pore properties are estimated by the  $\text{N}_2$  adsorption/desorption isotherm, and the results are shown in Figure 10b. Although the  $\text{N}_2$  adsorption/desorption isotherm of the Au foam hohlraum does not follow any particular pattern, due to some carbon contamination, it also exhibits an enhanced surface area of  $57.14 \text{ m}^2/\text{g}$  from the multi-point BET test. It is much higher than the Au nanowires ( $6.9 \text{ m}^2/\text{g}$ ) prepared by Ji et al., the Au foam (BET surface area:  $10.9 \text{ m}^2/\text{g}$ ) obtained from the combustion synthesis, and higher than the Au foam (BET surface area:  $23.5 \text{ m}^2/\text{g}$ ) fabricated using the PS beads as the template. It mainly results from the ultralow density of the Au foam hohlraum.

Furthermore, the Au foam hohlraum also possesses a high pore volume of  $0.11 \text{ cm}^3/\text{g}$ , which is close to the result using the PS beads as templates ( $0.10$  and  $0.15 \text{ cm}^3/\text{g}$ ). The pore size distribution was calculated from the desorption branch using the DFT method. It shows that the Au foam hohlraum has an average pore diameter of about  $38 \text{ nm}$  with a wide range distribution from  $25$  to  $360 \text{ nm}$ , and the result is consistent with the ligament size distribution.

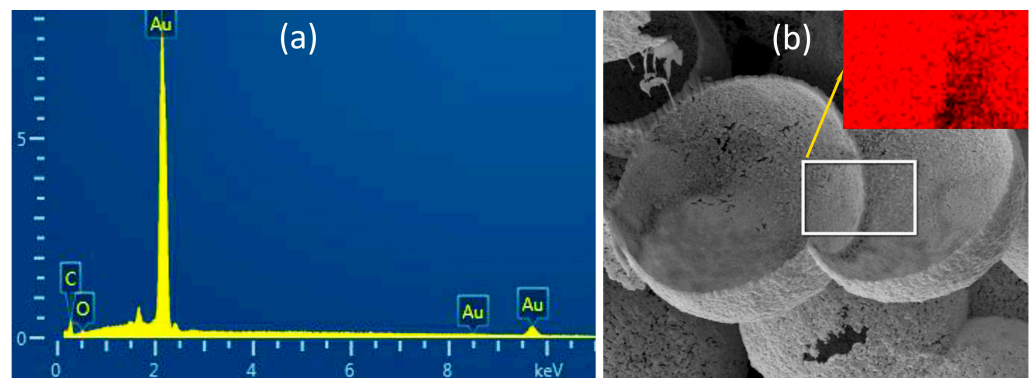


**Figure 10.** (a) XRD pattern and (b) pore size distribution of the Au foam hohlraum, the inset of (b) shows the nitrogen adsorption/desorption isotherm of the Au foam hohlraum.

Achieving Au foam with a density as low as  $0.28 \text{ g}/\text{cm}^3$  was possible by a template-assisted method using the PS beads as a template [35]. However, the Au foam fabricated by this method contained a large amount of carbon impurity, with a minimum content of 4%. It may result from the incomplete pyrolysis of the PS template in the  $\text{N}_2$  atmosphere among the temperature ranging from  $283$  to  $398^\circ\text{C}$ , only with a 96% theoretical weight loss of PS in this temperature region [28], as well as the alloying reaction of the Au and Ag, which takes place during this heat treatment. The existence of carbon as an impurity will decrease the conversion efficiency of the laser-X-ray due to the low atomic number [36]. The  $\text{SiO}_2$  microspheres were employed as templates to fabricate an Au foam to decrease the impurity and enhance the laser-X-ray conversion efficiency. High-purity Au foam was achieved via sintering the Ag-Au alloy, followed by the selected removal of the  $\text{SiO}_2$  template. The purity of the fabricated Au foam hohlraum was analyzed by the EDS, ICP-MS, and carbon analyzer, and the results are shown in Figure 11 and Table 1. According to the EDS spectrum, the vast majority of the detected element is gold, only a trace of carbon and oxygen components, and no Si or Ag are observed, indicating the complete removal of the Ag and  $\text{SiO}_2$  template. Carbon and oxygen are also confirmed by the carbon analyzer measurement, with a low concentration of  $0.25 \text{ wt}\%$  and  $0.38 \text{ wt}\%$ , respectively. The trace of C and O may result from the sample being exposed to the air atmosphere and adsorbing the  $\text{CO}_2$ . Interestingly, less than  $50 \text{ ppm}$  of Si and Ag are detected by the ICP-MS method, although it holds a higher sensitivity and lower detecting limitation. On the one hand, the  $\text{SiO}_2$  templates have a high melting point and exhibit excellent stability when the heat treatment temperature is below the melting point. On the other hand, the flowing  $\text{N}_2$  during the calcination provides an inert atmosphere and guarantees no side-reaction occurring for the Ag component. Moreover, the  $\text{SiO}_2$  templates and the Ag component can be fully corroded by different acid solutions, resulting in a high-purity Au foam hohlraum.

The fabricated Au foam hohlraum has been successfully applied to the physical experiment on the Shenguang III laser facility, which is used to verify the foam's gold suppression of plasma expansion and the enhancement of the re-emission energy flow in the hohlraum [37,38]. Compared to the Au block material with a density of  $19.3 \text{ g}/\text{cm}^3$ , the velocities of the X-ray emission fronts moving off the wall are much smaller than the Au foam hohlraum with a density of  $0.3 \text{ g}/\text{cm}^3$  [39]. The XFC result indicates that the X-ray

re-emissions from the Au foam hohlraum are enhanced by about 10.5%, between 1.1 and 1.5 ns, and the TGS measurement demonstrates that the time-integrated emission increases by about 12%. In addition, the increasing emission fraction by the Au foam increases with time [37,40]. Thus, it is believed that the ultralow-density, high-purity Au foam hohlraum with hierarchical porous structures synthesized by the template-dealloying method have advantages in symmetry controlling and lowering the plasma filling, which will have profound application in ICF high energy-density physical experiments.



**Figure 11.** (a) EDS spectrum and (b) cross image of the Au foam hohlraum, the inset of (b) shows the corresponding EDS elemental mapping image.

**Table 1.** Elemental composition of pure Au foam hohlraum.

Test Method	Si	C	O	Ag	Au
EDS	—	✓	✓	—	✓
ICP-MS	<50 ppm	—	—	<50 ppm	—
Carbon Analyzer	—	0.25 wt%	0.38 wt%	—	—

#### 4. Conclusions

In conclusion, we developed a new process to prepare ultralow-density, high-purity Au foam with a hierarchical porous structure using the template-dealloying method, using SiO<sub>2</sub> beads as the sacrificial template. An Au foam hohlraum with low-density (0.20 g/cm<sup>3</sup>) and high-purity (99.37%) was obtained. This novel method not only separates the interaction between the sintering molding and template removal but also reduces the impurity content of the Au foam significantly. Moreover, it also decreases the radial shrinkage of the hollow microsphere, which benefits further lowering the density of the Au foam. By using the self-designed hohlraum forming mold, Au foam hohlraum, with different heights and different thicknesses of low-density layers, was obtained by adjusting the mode size. Moreover, the Au foam hohlraum samples fully retained the microstructure of the ultralow-density Au foam. The Au foam hohlraum consisted of an abundant free-standing single Au hollow shell composed of a homogeneous nanoporous structure.

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